

# Rapid Computation of thermal stress in crystals with facets and allowing for material anisotropy

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Joint work with  
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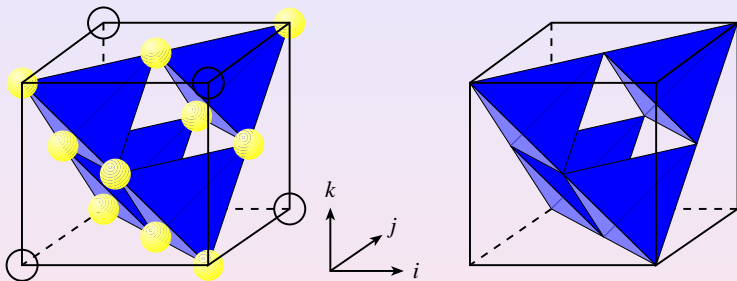
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# Coordination Polyhedra

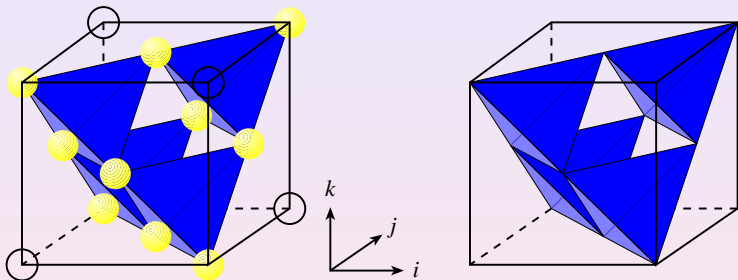
- The growth rate is based on a coordinate polyhedron model
- This is capable of naturally explaining the different growth rates between the positive and negative directions in a polar crystal such as the III-V semiconductors
- If AB is the III-V semiconductor under consideration, then its anion-coordination polyhedra are  $AB_4^{6-}$  tetrahedra

# Coordination Polyhedra



Shown are the four tetrahedra of an AB unit cell. To the left only the B atoms in the unit cell are shown. B atoms in the unit cell but not included in the four growth units are represented with hollow circles. At the centre of each tetrahedral growth unit is a A atom accounting for all the atoms in the AB unit cell. On the right only the tetrahedra are shown.

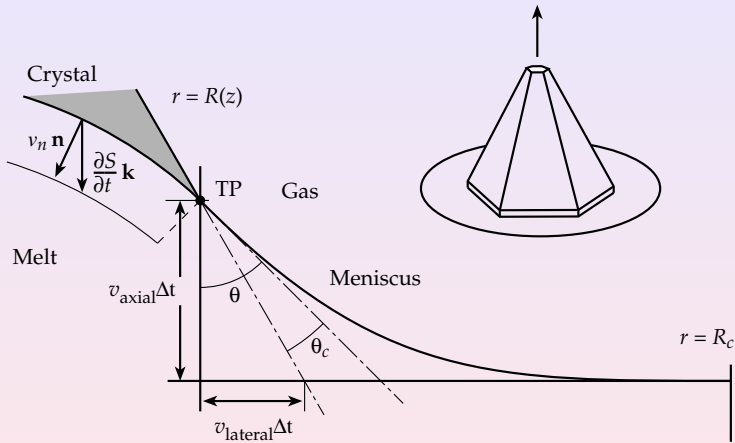
# Coordination Polyhedra



For a crystal pulled in the  $[001]$  direction,

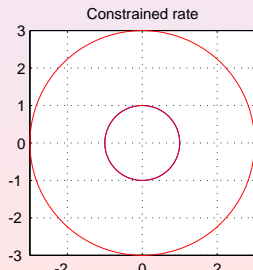
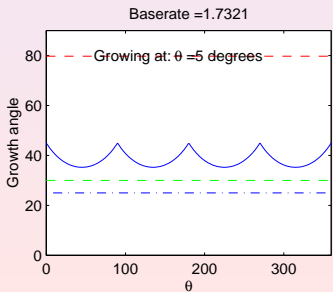
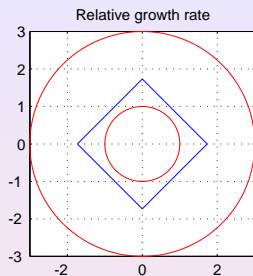
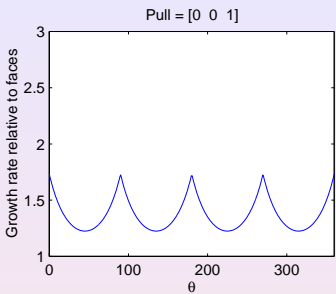
- $[00\bar{1}]$  is into the melt gives  $v_{\text{axial}} = 1.7321$
- $v_{\text{lateral}}$  has four-fold symmetry

# Constrained Growth

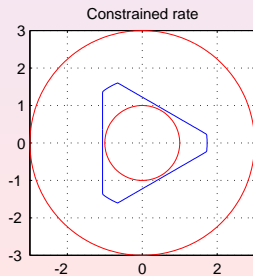
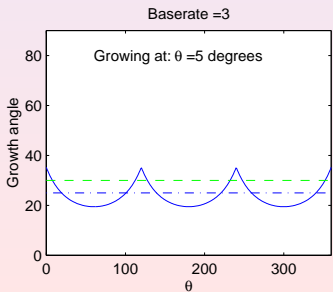
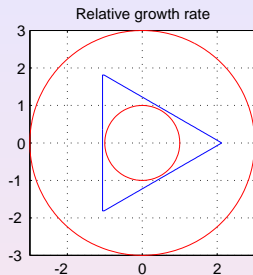
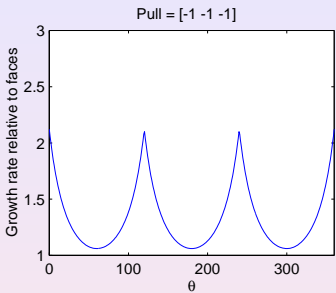


- If not constrained by the meniscus then  $\tan(\theta - \theta_c) = \frac{v_{lateral}}{v_{axial}}$
- For growing a cone  $\theta - \theta_c$  is 1/2 the opening angle of the cone

# Pulling in the [001] direction

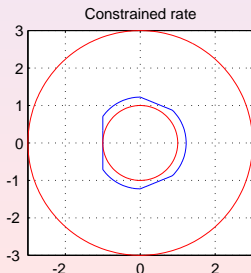
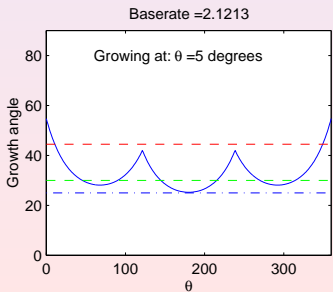
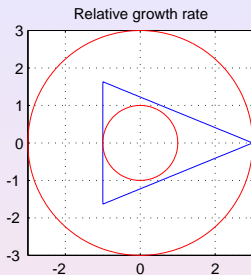
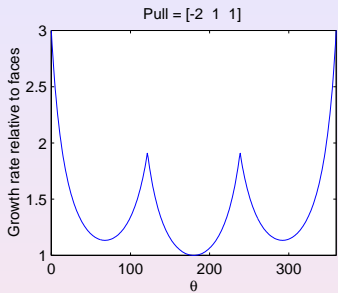


# Pulling in the $[\bar{1}\bar{1}\bar{1}]$ direction

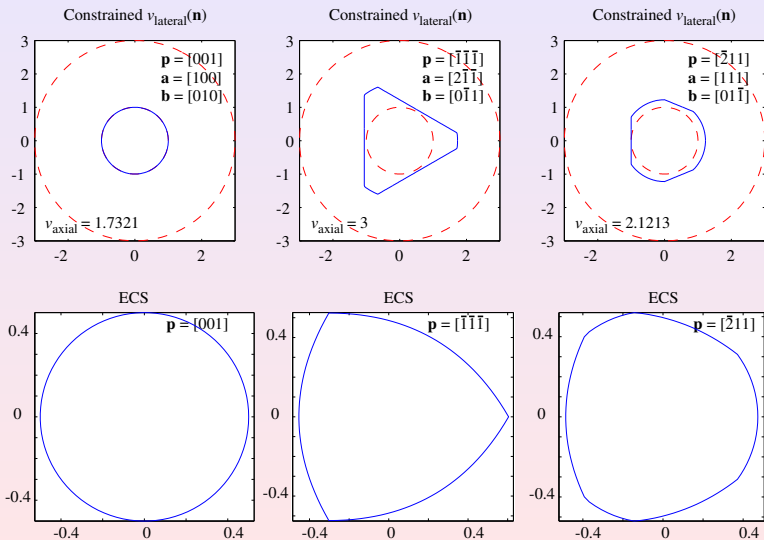




# Pulling in the $[\bar{2}11]$ direction



# Equilibrium Crystal Shapes



# Equilibrium Crystal Shapes

For the purpose of computing thermal stress, we assume the following expression in the case of weak anisotropy ( $\alpha$  small)

$$R(\phi, z) = \bar{R}(z) \left( 1 + \alpha \sum_{k=1}^m \beta_k \cos(n_k \phi + \delta_k) \right),$$

where  $m, n_1 < n_2 < \dots < n_m$  are positive integers and  $\sum_{k=1}^m \beta_k^2 = 1$ .

- $\alpha$  is the (small) geometric anisotropy factor
- 4-fold symmetry ( $m = 1, n_1 = 4$ )
- 6-fold symmetry ( $m = 1, n_1 = 6$ )
- We assume that the lateral shape of the crystal is in equilibrium

# Basic Equations

Within the crystal  $\Omega$ , the temperature  $T(\mathbf{x}, t)$  satisfies the heat equation,

$$\rho_s c_s \frac{\partial T}{\partial t} = \nabla \cdot (\kappa_s \nabla T), \quad \mathbf{x} \in \Omega, \quad t > 0$$

where  $\rho_s$ ,  $c_s$  and  $k_s$  are the density, specific heat, and thermal conductivity of the crystal. The boundary conditions are below,

$$\begin{aligned} -\kappa_s \frac{\partial T}{\partial \mathbf{n}} &= h_{gs}(T - T_g) + h_F(T^4 - T_b^4), & \mathbf{x} \in \Gamma_g, \\ \kappa_s \frac{\partial T}{\partial z} &= h_{ch}(T - T_{ch}), & z = 0, \end{aligned}$$

where  $h_{gs}$  and  $h_{ch}$  represent the heat transfer coefficients;  $h_F$  the radiation heat transfer coefficient;  $T_g$ ,  $T_{ch}$  and  $T_b$  denote the ambient gas temperature, the chuck temperature and background temperature respectively.

# Basic Equations

The crystal/melt interface is denoted  $\Gamma_S$  and is where  $T = T_m$ , the melting temperature. Explicitly we denote the melting isotherm by

$$z - S(\mathbf{x}, t) = 0, \quad \mathbf{x} \in \Gamma_S.$$

The motion of the interface of the phase transition is governed by the Stefan condition

$$\rho_S L |\mathbf{v}_n| = \kappa_S \left. \frac{\partial T}{\partial \mathbf{n}} \right|_{z \rightarrow S^-} - q_{l,n}, \quad |\mathbf{v}_n| = v_n = \frac{\partial S}{\partial t} \mathbf{k} \cdot \mathbf{n}$$

where  $L$  is the latent heat,  $|\mathbf{v}_n|$  is the speed of the interface in the direction of its outward normal  $\mathbf{n}$ , and  $q_{l,n}$  is the heat flux from the melt normal to the interface. The speed  $\partial S / \partial t$  is the speed of the interface  $S$  in the  $\mathbf{k}$  direction.

# Rescaled Equations

Identify the Biot number

$$\epsilon = \frac{\bar{h}_{gs} \tilde{R}}{\kappa_s} \quad (1)$$

as a small parameter (small lateral heat flux). Rescaling,

$$\frac{\epsilon}{St} \Theta_t = \frac{1}{r} (r \Theta_r)_r + \frac{1}{r^2} \Theta_{\phi\phi} + \epsilon \Theta_{zz}, \quad \mathbf{x} \in \Omega, t > 0,$$

with,

$$-\Theta_r + \frac{1}{R^2} R_\phi \Theta_\phi + \epsilon R_z \Theta_z = \epsilon F(\Theta) \left( 1 + \frac{R_\phi^2}{R^2} + \epsilon R_z^2 \right)^{1/2}, \quad \mathbf{x} \in \Gamma_g,$$

$$\Theta_z(0, \phi, t) = \delta (\Theta(0, \phi, t) - \Theta_{ch}),$$

$$\Theta = 1, \quad \mathbf{x} \in \Gamma_s,$$

$$\Theta_z - \frac{1}{\epsilon} S_r \Theta_r - \frac{1}{\epsilon r^2} S_\phi \Theta_\phi = \gamma + S_t, \quad \gamma = \frac{q_l \tilde{R}}{\epsilon^{1/2} \kappa_s \Delta T}.$$

# Rescaled Equations

$\beta(z) = h_{gs}/\bar{h}_{gs}$ , and  $\delta = \epsilon^{1/2} h_{ch}/\bar{h}_{gs}$  and  $\gamma(q_l)$  is the non-dimensional (dimensional) heat flux in the liquid across the crystal/melt interface in the axial direction. Also,

$$F(\Theta) = \frac{h_F(T_g^4 - T_b^4)}{\bar{h}_{gs}\Delta T} + \left( \beta(z) + \frac{4h_F}{\bar{h}_{gs}} T_g^3 \right) \Theta + \frac{h_F}{\bar{h}_{gs}} \Delta T (6T_g^2 + 4T_g\Delta T\Theta + \Delta T^2\Theta^2) \Theta^2.$$

# Perturbation Solution

The Biot number for the lateral heat flux is small ( $\epsilon \sim 0.03$ ) and the geometric anisotropy is weak ( $\alpha \ll 1$ ).

Expansion:

$$\begin{aligned}\Theta &\sim \Theta_0(z, t) + \epsilon \Theta_1(r, \phi, z, t) + \epsilon^2 \Theta_2(r, \phi, z, t) + \dots, \\ S &\sim S_0(t) + \epsilon S_1(r, \phi, t) + \epsilon^2 S_2(r, \phi, t) + \dots.\end{aligned}$$

Zeroth order model (**Fast to compute**):

$$\begin{aligned}\frac{1}{St} \Theta_{0,t} - \Theta_{0,zz} &= \frac{2}{\bar{R}} (\bar{R}' \Theta_{0,z} - F(\Theta_0)), & 0 < z < S_0(t), & t > 0, \\ \Theta_{0,z}(0, t) &= \delta(\Theta_0(0, t) - \Theta_{ch}), & & t \geq 0, \\ \Theta_0(S_0(t), t) &= 1, & & t \geq 0, \\ S_0'(t) &= \Theta_{0,z}(S_0(t), t) - \gamma, & S_0(0) = Z_0, & t > 0.\end{aligned}$$



# Perturbation Solution

First order model:

$$\Theta_1(r, \phi, z, t) = \Theta_1^a(z, t) + r^2\Theta_1^b(z, t) + \alpha\Theta_1^c(r, \phi, z, t) + O(\alpha^2)$$

where, keeping only those terms to  $O(\alpha)$ ,

$$\Theta_1^b(z, t) = \frac{1}{2\bar{R}} (\bar{R}'\Theta_{0,z} - F(\Theta_0)),$$

$$\Theta_1^c(r, \phi, z, t) = \bar{R}F(\Theta_0) \sum_{k=1}^m \frac{\beta_k}{n_k} \left(\frac{r}{\bar{R}}\right)^{n_k} \cos(n_k\phi + \delta_k).$$

These last two terms are completely determined by  $\Theta_0$  and  $\bar{R}$ .  $\Theta_1^a$  does not play a role in the stress.

# Basic Relations

For a crystal with cubic symmetry the stresses

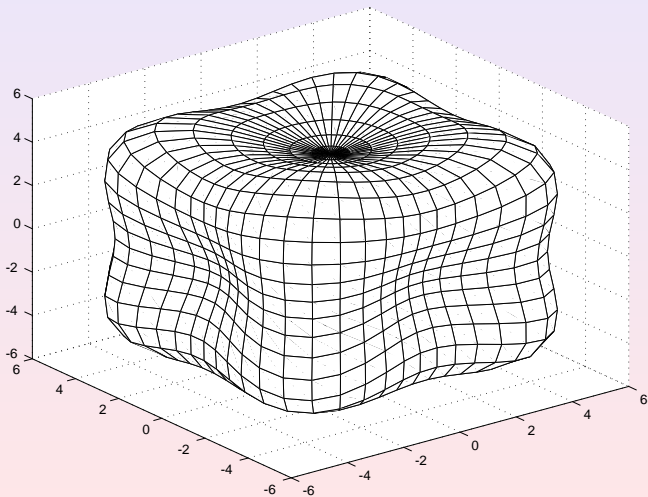
$\underline{\sigma} = (\sigma_{xx}, \sigma_{yy}, \sigma_{zz}, \sigma_{yz}, \sigma_{xz}, \sigma_{xy})^T$  and strains

$\underline{e} = (e_{xx}, e_{yy}, e_{zz}, 2e_{yz}, 2e_{xz}, 2e_{xy})^T$  are related through

$$\underline{\sigma} = C_{\text{rect}} \underline{e}, \quad C_{\text{rect}} = \begin{pmatrix} C_{11} & C_{12} & C_{12} & & & \\ C_{12} & C_{11} & C_{12} & & & \\ C_{12} & C_{12} & C_{11} & & & \\ & & & C_{44} & & \\ & & & & C_{44} & \\ & & & & & C_{44} \end{pmatrix}.$$

For an anisotropic material the quantity  $H = 2C_{44} - C_{11} + C_{12} \neq 0$ . We assume that the z-component of the displacement is zero because of the free surface at the melt.

# Directional Dependence of the Young's modulus for an INSB Crystal



# Operator Splitting

Split  $C_{\text{rect}}$  into a diagonal anisotropic part and an isotropic part

$C_{\text{rect}} = C_0 - C_{a,\text{rect}}$ ,  $C_{a,\text{rect}} = H/4 \times \text{diag}(2, 2, 2, -1, -1, -1)$ , and

$$C_0 = \begin{pmatrix} C_{11}^0 & C_{12}^0 & C_{12}^0 & & & \\ C_{12}^0 & C_{11}^0 & C_{12}^0 & & & \\ C_{12}^0 & C_{12}^0 & C_{11}^0 & & & \\ & & & C_{44}^0 & & \\ & & & & C_{44}^0 & \\ & & & & & C_{44}^0 \end{pmatrix}$$

is isotropic.  $C_{a,\text{rect}}$  is chosen to minimize  $\rho(C_0^{-1} C_{a,\text{rect}})$ .

$E$  and  $\nu$  in term of  $C_{ij}$  are given by

$$E = \frac{(C_{11} + 2C_{12} + H/2)(C_{11} - C_{12} + H/2)}{C_{11} + C_{12} + H/2},$$

$$\nu = \frac{C_{12}}{C_{11} + C_{12} + H/2}.$$

# Operator Splitting

Denote the displacement vector as  $\mathbf{w}$ , the strain by  $\mathbf{e} = \mathbf{S}(\mathbf{w})$  and the stress by  $\sigma = C\mathbf{S}(\mathbf{w})$  with  $C = C_0 - C_a$ .

The thermoelastic problem becomes

$$\begin{aligned}\nabla \cdot C\mathbf{S} &= (C_{11} + 2C_{12})\nabla\Theta, & \mathbf{x} \in \Omega, \quad t > 0, \\ C\mathbf{S} \cdot \mathbf{n} &= (C_{11} + 2C_{12})\Theta\mathbf{n}, & r = R(\phi, z)\end{aligned}$$

or by rescaling

$$\begin{aligned}\nabla \cdot C\mathbf{S} &= \left( \frac{1-\nu}{1-2\nu} - \frac{H}{2} \right) \nabla\Theta, & \mathbf{x} \in \Omega, \quad t > 0, \\ C\mathbf{S} \cdot \mathbf{n} &= \left( \frac{1-\nu}{1-2\nu} - \frac{H}{2} \right) \Theta\mathbf{n}, & r = R(\phi, z)\end{aligned}$$

with  $\mathbf{n}$  denoting the outward normal of the surface  $r = R(\phi, z)$ .

# Operator Splitting

Using the form of  $C$ ,

$$\begin{aligned}\nabla \cdot C\mathbf{S} &= \nabla \cdot C_0\mathbf{S} - \nabla \cdot C_a\mathbf{S} = \mathcal{L}_0 - \mathcal{L}_a, \\ C\mathbf{S} \cdot \mathbf{n} &= C_0\mathbf{S} \cdot \mathbf{n} - C_a\mathbf{S} \cdot \mathbf{n} = \mathcal{B}_0 - \mathcal{B}_a,\end{aligned}$$

to solve for  $\mathbf{w}(\mathbf{x})$  one starts with  $\mathbf{w}_0$  given by

$$\begin{aligned}\mathcal{L}_0(\mathbf{w}_0) &= \left( \frac{1-\nu}{1-2\nu} - \frac{H}{2} \right) \nabla \Theta, & \mathbf{x} \in \Omega, \quad t > 0, \\ \mathcal{B}_0(\mathbf{w}_0) &= \left( \frac{1-\nu}{1-2\nu} - \frac{H}{2} \right) \Theta \mathbf{n}, & r = R(\phi, z).\end{aligned}$$

$\mathbf{w}_0$  is the isotropic displacement found previously [Bohun et al.], multiplied by a factor of  $1 - \frac{H}{2} \frac{1-2\nu}{1-\nu}$ .

# Operator Splitting

We know  $\mathbf{w}_0$  explicitly for a given crystal shape  $R(\phi, z)$ .

Having defined  $\mathbf{w}_0$ , we denote by  $\mathbf{w}_{k+1} = \mathcal{N}\mathbf{w}_k$ , with  $k \geq 0$ , the solution to

$$\begin{aligned}\mathcal{L}_0(\mathbf{w}_{k+1}) &= \mathcal{L}_a(\mathbf{w}_k), & \mathbf{x} \in \Omega, \quad t > 0, \\ \mathcal{B}_0(\mathbf{w}_{k+1}) &= \mathcal{B}_a(\mathbf{w}_k), & r = R(\phi, z).\end{aligned}$$

# Perturbation Series

Continuing this process we have for  $\mathbf{w}(\mathbf{x})$

$$\mathbf{w} = \mathbf{w}_0 + \mathcal{N}\mathbf{w}_0 + \mathcal{N}^2\mathbf{w}_0 + \cdots + \mathcal{N}^n\mathbf{w}_0 + \cdots .$$

Since  $\|\mathcal{N}\| \leq \omega$  in a suitable norm, where

$$\omega = \frac{|H|/2}{C_{11} - C_{12} + H/2} = \frac{|2C_{44} - C_{11} + C_{12}|}{2C_{44} + C_{11} - C_{12}} < 1$$

is an anisotropic factor, the series converges and an error can be estimated when replaced by a finite sum. For typical cubic anisotropic materials  $\omega \sim 1/3$ .

	$C_{11}$	$C_{12}$	$C_{44}$	$\omega$
GAAs	$12.16 \times 10^4$	$5.43 \times 10^4$	$6.18 \times 10^4$	0.295
INP	$10.76 \times 10^4$	$6.08 \times 10^4$	$4.233 \times 10^4$	0.288
INSB	$6.70 \times 10^4$	$3.65 \times 10^4$	$3.02 \times 10^4$	0.329



# Perturbation Series

- For a given pulling direction  $C_0$  is invariant however, the explicit form of  $C_a$  depends on the crystal orientation
- Consequently  $\mathcal{L}_a$  and  $\mathcal{B}_a$  depend on the orientation
- $C_a$  transforms as a fourth rank tensor and includes only trigonometric factors  $\cos m\phi$  and  $\sin m\phi$  where  $m$  depends on the orientation of the crystal

For example, if  $(c_4, s_4) = (\cos 4\phi, \sin 4\phi)$  then

$$C_{a,cyc}^{[001]} = \frac{H}{4} \begin{pmatrix} 1 + c_4 & 1 - c_4 & 0 & 0 & 0 & -s_4 \\ 1 - c_4 & 1 + c_4 & 0 & 0 & 0 & s_4 \\ 0 & 0 & 2 & 0 & 0 & 0 \\ 0 & 0 & 0 & -1 & 0 & 0 \\ 0 & 0 & 0 & 0 & -1 & 0 \\ -s_4 & s_4 & 0 & 0 & 0 & -c_4 \end{pmatrix}.$$

# Plane Strain

To illustrate the procedure assume that the displacement is only in the  $(r, \phi)$  plane.

Stress strain relation for the  $[001]$  direction becomes

$$\begin{pmatrix} \sigma_{a,rr} \\ \sigma_{a,\phi\phi} \\ \sigma_{a,r\phi} \end{pmatrix} = \frac{H}{4} \begin{pmatrix} 1 + c_4 & 1 - c_4 & -s_4 \\ 1 - c_4 & 1 + c_4 & s_4 \\ -s_4 & s_4 & -c_4 \end{pmatrix} \begin{pmatrix} e_{rr} \\ e_{\phi\phi} \\ 2e_{r\phi} \end{pmatrix}.$$

For the  $[\bar{1}\bar{1}\bar{1}]$  direction

$$\begin{pmatrix} \sigma_{a,rr} \\ \sigma_{a,\phi\phi} \\ \sigma_{a,r\phi} \end{pmatrix} = \frac{H}{12} \begin{pmatrix} 0 & 2 & 0 \\ 2 & 0 & 0 \\ 0 & 0 & -1 \end{pmatrix} \begin{pmatrix} e_{rr} \\ e_{\phi\phi} \\ 2e_{r\phi} \end{pmatrix}.$$

# A Canonical Problem

To find  $\mathbf{w}_0 + \mathbf{w}_1 = \mathbf{w}_0 + \mathcal{N}\mathbf{w}_0$  the thermoelastic equations

$$\begin{aligned}\mathcal{L}_0(\mathbf{w}_1) &= \mathcal{L}_a(\mathbf{w}_0), & \mathbf{x} \in \Omega, \quad t > 0, \\ \mathcal{B}_0(\mathbf{w}_1) &= \mathcal{B}_a(\mathbf{w}_0), & r = R(\phi, z)\end{aligned}$$

reduce to finding sequence of solutions of the form

$$\begin{aligned}\frac{\partial \sigma_{rr}}{\partial r} + \frac{1}{r} \frac{\partial \sigma_{r\phi}}{\partial \phi} + \frac{\sigma_{rr} - \sigma_{\phi\phi}}{r} &= f_r r^{k-2} \cos(n\phi + \delta), \quad r < \bar{R}(z), \\ \frac{\partial \sigma_{r\phi}}{\partial r} + \frac{1}{r} \frac{\partial \sigma_{\phi\phi}}{\partial \phi} + \frac{2\sigma_{r\phi}}{r} &= f_\phi r^{k-2} \sin(n\phi + \delta), \quad r < \bar{R}(z),\end{aligned}$$

with integers  $n \geq 0$ ,  $k \geq 1$ , and

$$\begin{aligned}\sigma_{rr} &= g_r r^{k-1} \cos(n\phi + \delta), & r = \bar{R}(z), \\ \sigma_{r\phi} &= g_\phi r^{k-1} \sin(n\phi + \delta), & r = \bar{R}(z),\end{aligned}$$

where  $f_r, f_\phi, g_r, g_\phi$  depend on  $C_a$ .

# A Canonical Problem

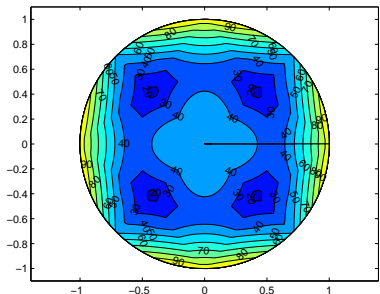
We solve this with a two stage approach.

- 1 Find a particular solution that does not necessarily satisfy the boundary condition
- 2 Find a homogeneous solution with a (perhaps) modified boundary condition

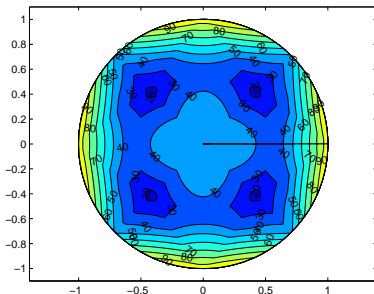
The point here is that the solution can be written out explicitly for general  $f_r, f_\phi, g_r, g_\phi$  so that the problem becomes a bookkeeping problem.

Fast

# Results - Geometric [001]: Total Resolved Stress

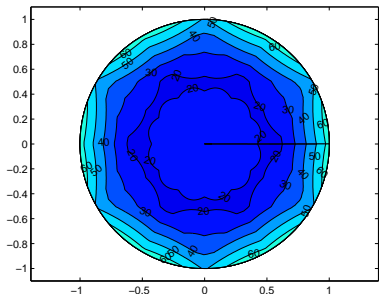


$$\alpha = 0: \max |\sigma_{rs}^{\text{tot}}| = 9.23$$

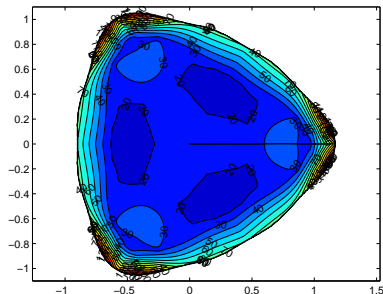


$$\alpha = 0: \max |\sigma_{rs}^{\text{tot}}| = 9.23$$

# Results - Geometric $[\bar{1}\bar{1}\bar{1}]$ : Total Resolved Stress

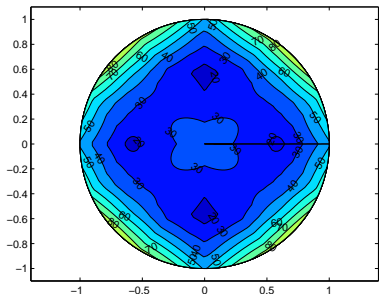


$\alpha = 0: \max |\sigma_{rs}^{tot}| = 6.07$

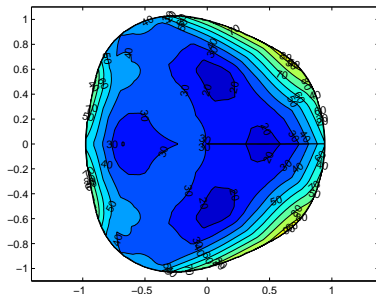


$\alpha = 0.123: \max |\sigma_{rs}^{tot}| = 13.4$

# Results - Geometric $[\bar{2}11]$ : Total Resolved Stress

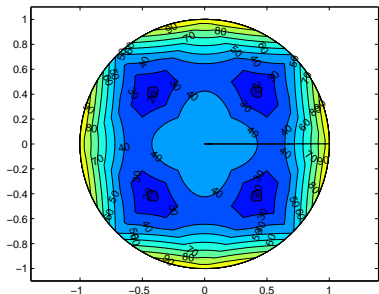


$\alpha = 0: \max |\sigma_{rs}^{tot}| = 8.19$

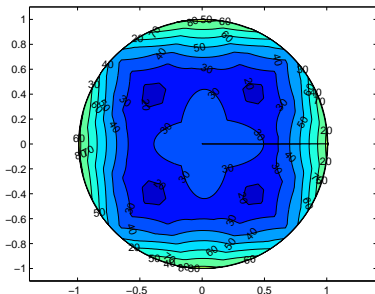


$\alpha = 0.089: \max |\sigma_{rs}^{tot}| = 8.78$

# Results - Anisotropy [001]: Total Resolved Stress



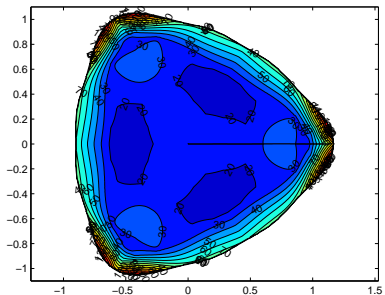
$$\omega = 0: \max |\sigma_{rs}^{\text{tot}}| = 9.23$$



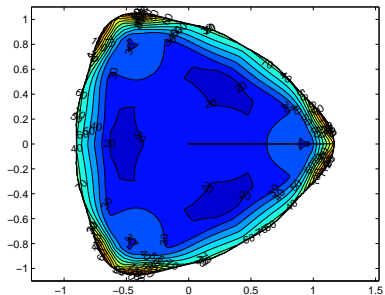
$$\omega = 0.329: \max |\sigma_{rs}^{\text{tot}}| = 7.66$$



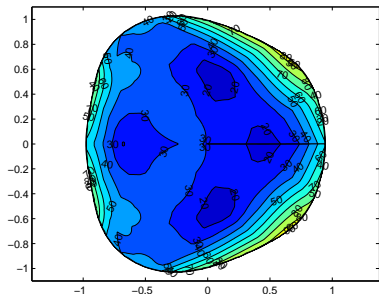
# Results - Anisotropy $[\bar{1}\bar{1}\bar{1}]$ : Total Resolved Stress



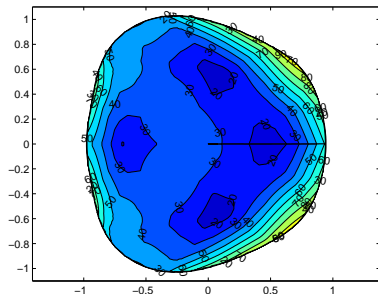
$\omega = 0: \max |\sigma_{rs}^{\text{tot}}| = 13.4$



$\omega = 0.329: \max |\sigma_{rs}^{\text{tot}}| = 12.0$

Results - Anisotropy  $[\bar{2}11]$ : Total Resolved Stress

$$\omega = 0: \max |\sigma_{rs}^{\text{tot}}| = 8.19$$



$$\omega = 0.329: \max |\sigma_{rs}^{\text{tot}}| = 8.78$$

# Conclusions

- A simple argument based on the crystal lattice structure predicts facets that depend on both the crystal orientation and growth angle
- Small opening angles tend to suppress the formation of facets
- The model naturally incorporates the polarity of III-V semiconductors
- Facet formation greatly affects the thermal stress distribution
- Anisotropy has a lesser effect when the crystal has facets
- The industry preference of the  $[\bar{2}11]$  pulling direction, determined by trial and error, produces facets yet avoids the drastic increase in the stress seen in the  $[\bar{1}\bar{1}\bar{1}]$  orientation. Furthermore, effect of the material anisotropy is negligible in this case

Thank you